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ECR Ar/O₂ Plasma Oxidation of HfN Thin Films for High Dielectric HfO_xN_y Formations

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Electron cyclotron resonance (ECR) plasma oxidation of HfN thin films was investigated for high dielectric HfO_xN_y thin films formations. After high vacuum annealing (HVA) at 700°C for 3 min followed by 1000°C post-deposition annealing (PDA) for 1 min in flowing N₂ ambient, equivalent oxide thickness (EOT) of 1.3 nm with the leakage current of $4.3 \times 10^{-3} \text{ A/cm}^2$ (@V_{FB}-1V) was obtained, however, the V_{FB} shift still remained. It was found that the electrical characteristics were dramatically improved after 700°C HVA for 5 min followed by 1000°C PDA for 2 min for EOT of 1.4 nm with a leakage current of 2.3 x 10⁻³ A/cm² (@V_{FB}-1V) and a V_{FB} shift of -0.05 V.

Key words: HfON, ECR, high-k, gate dielectric, plasma oxidation

1. INTRODUCTION

SiO₂ gate dielectric has become thiner than 2 nm or less with scaling of MOSFET. It causes the increase of gate leakage current by direct tunneling conduction. Therefore, high-k gate dielectrics, such as ZrO₂, HfO₂, and La₂O₃, are being investigated as alternative gate insulators [1]. Among these materials, HfO2 is one of the more promising materials due to its high dielectric constant (k=25) and suitable barrier heights for electrons and holes in terms of the gate insulator applications. However, HfO₂ has some problems such as crystallization by high temperature annealing, formation of an interfacial layer with a low dielectric constant, flat-band voltage (V_{FB}) shift, and boron penetration from p⁺-poly Si gate. On the other hand, HfO_xN_y thin films show some advantages such as good thermal stability, suppression of the interfacial layer formation, and resistance to boron penetration [2, 3].

In this paper, HfO_xN_y high-k thin films formed by the electron cyclotron resonance (ECR) plasma oxidation of HfN thin films deposited on p/p^+ -Si(100) by the ECR sputtering method are investigated. This method would be expected to form high-quality gate insulator thin films because the ECR sputtering causes little damage to the deposited films, and HfN thin films would be expected to suppress interfacial layer formation during Ar/O₂ plasma oxidation and post-deposition annealing (PDA) processes [4, 5].

2. EXPERIMENTAL PROCEDURE

The p/p^+ -Si(100) wafers were chemically cleaned in H_2SO_4/H_2O_2 and dipped in diluted HF followed by a rinse in ultra-pure water (Organo UPW system). HfN thin films (1 nm) were deposited on p/p^+ -Si(100) at room temperature by ECR plasma sputtering. The pressure in the chamber during the deposition was 9 x 10^{-2} Pa with Ar/N₂ flow rates of 20/8 sccm. The HfN thin films were then oxidized by ECR Ar/O₂ plasma for 60 s to form the HfO_xN_y thin films with Ar/O₂ flow rates of 20/8 sccm. The deposited HfO_xN_y films were

annealed in a vacuum (10⁻⁴ Pa) at 700°C for 3 or 5 min (HVA). The post-deposition annealing (PDA) was also carried out by the rapid thermal annealing system (ULVAC VHC-P610) at 400 to 1000°C for 1 to 10 min in the flowing N₂ ambient. Finally, Al electrode was deposited evaporation. The by fabricated Al/HfO_xN_y/p-Si(100) MOS diodes were characterized capacitance-voltage (C-V) bv and current density-voltage (J-V) measurements. C-V was measured in the frequency range of 10 k - 1 MHz by LCR meter (HP 4284A), and J-V was measured by semiconductor parameter analyzer (HP 4156A). The equivalent oxide thickness (EOT) was calculated from the obtained C-V results by using exponential potential based quantum mechanical extraction method (EPOQUE) [6]. The surface morphology of the HfO_xN_y films was observed by tapping mode AFM (Atomic Force Microscopy, Digital Instrument : Nanoscope III). XPS (X-ray Photoelectron Spectroscopy) measurement was also performed by using Al Ka non-monochromatic X-ray source.

3. RESULTS AND DISCUSSION

3.1 Effect of Post-Deposition Anneal

Figure 1 shows the AFM images for HfO_xN_y films formed by ECR plasma oxidation of the HfN thin films followed by the HVA for 3 min. The surface morphologies were found to be very smooth for the films both before and after the 1000°C PDA. The obtained rms surface roughness was approximately 0.14 nm.

Figure 2 shows PDA temperature dependence of C-V (1 MHz) and J-V characteristics for the HfO_xN_y thin films. As shown in Fig. 2(a), the V_{FB} shift was quite large in case of without PDA, while it was improved with the increase of PDA temperature. After the 1000°C PDA, the V_{FB} shift was improved to -0.44 V. Interestingly, the equivalent oxide thicknesses (EOTs) became thin with the increase of PDA temperature, and an EOT of 1.3 nm was obtained and the hysteresis width



Fig. 1 AFM images for HfO_xN_y thin films(x, y: 1µm x 1µm, z: 10 nm/div.).(a) After HVA for 3 min(w/o PDA) (rms: 0.14 nm),and(b) after 1000°C PDA (rms: 0.14nm).



Fig. 2. PDA temperature dependence of (a) C-V (1 MHz) and (b) J-V characteristics for the HfO_xN_y thin films.

was also improved from 80 mV down to 3 mV after the 1000°C PDA. Furthermore, the leakage current density was decreased after the PDA. By performing 1000°C PDA, the leakage current was decreased to 4.3×10^{-3}



Fig. 3. Frequency dependence of C-V characteristics for the HfO_xN_y thin films, (a) without PDA, (b) 600°C PDA, (c) 1000°C PDA.

 A/cm^{2} (@ V_{FB}-1V).

The frequency dependence of C-V characteristic for the HfO_xN_y thin films is shown in Fig. 3. The severe frequency dispersion observed in the C-V for the film without PDA (Fig. 3(a)) was significantly improved by performing the PDA. After 600°C PDA, the V_{FB} shifts with measurement frequency were clearly observed, although the hysteresis was negligible as shown in Fig. 3(b), while those were dramatically improved by the 1000°C PDA except the slight dispersion at the weak inversion region. The V_{FB} shift of -0.44 V was obtained as shown in Fig. 3(c).

Figure 4 shows the angle-resolved Si2p and Hf4f photoelectron spectra for the HfO_xN_y film without PDA. The take-off angles of XPS measurements were 30 to 80°. From the Si2p spectra, it was found that the Si-O



Fig. 4. Angle-resolved (a) Si2p and (b) Hf4f photoelectron spectra for HfO_xN_y films without PDA.

peaks increased with the take-off angles as shown in Fig. 4(a). This result suggests the oxygen atoms diffused well into HfO_xN_y films during the Ar/O_2 plasma oxidation and HVA processes. Furthermore, from the Hf4f spectra shown in Fig. 4(b), it was confirmed that the HfN thin films were completely oxidized by ECR Ar/O_2 plasma irradiation. In addition, the nitrogen atoms were found to diffuse toward the surface and the oxygen atoms toward the interface in HfO_xN_y films by performing ECR plasma oxidation and HVA. This result suggested that Hf-O binding is dominant at the interface region, while Hf-N binding increased at the surface region.

Figure 5 shows Si2p and Hf4f photoelectron spectra (take-off angle: 80°) for the HfO_xN_y/Si after the PDA. In this measurement condition, the binding configuration close to the interface of the HfO_xN_y/Si could be observed. As shown in Fig. 5(a), the Si2p spectra peak was chemically shifted to the Si-O binding energy and its intensity became strong after performing PDA with increasing PDA temperature. The Hf4f spectra peak was also shifted to the Hf-O binding energy, as shown in Fig. 5(b). These spectra shown in Fig. 5 led to the conclusion that the oxygen diffused toward the HfO_xN_y/Si interface during the PDA so that the HfSiON interface layer was formed after the PDA.

3.2 Time Dependence of Post-Deposition Anneal

As described in the previous section, HfO_xN_y films formed by ECR Ar/O₂ plasma oxidation of the HfN thin films showed excellent electrical characteristics such as



Fig. 5. Photoelectron spectra for HfO_xN_y /Si before and after the 1000°C PDA (take-off angle: 80°). (a) Si2p and (b) Hf4f.

small EOT with low leakage current and small hysteresis width. However, the negative V_{FB} shift still remained. In order to suppress the V_{FB} shift, the annealing time of HVA and PDA was investigated. First of all, the HVA was carried out for 5 min for improving the characteristics of deposited films, and the 1000°C PDA, which was the temperature showed the best electrical characteristics thus far, was further investigated to have the optimum condition with increasing PDA duration in the range of 1 to 10 min.

Figure 6 shows the PDA time dependence of the C-V (1 MHz) and J-V characteristics for HfO_xN_y thin films after the HVA for 5 min. As shown in Fig. 6(a), the V_{FB} shifts in C-V characteristics were almost suppressed for each PDA duration. It was considered that the HVA for 5 min led to a reduction of the positive fixed charge that existed in the HfO_xN_y thin films. It should be noted that the C-V characteristics and the V_{FB} shift for the film without PDAs was significantly improved compared to the film after the HVA for 3 min, as shown in Fig. 2(a) and Fig. 6(a), respectively. After 1000°C PDA for 3 min, the EOT of 1.3 nm with the leakage current of 7.5 x 10^{-3} A/cm² (@ V_{FB}-1V) was obtained for the HfO_xN_y film and the V_{FB} shift was approximately -0.02 V. The C-V characteristic of the HfOxNy film after the 1000°C PDA for 2 min was also excellent ; the EOT of 1.4 nm with the leakage current density of 2.3 x 10⁻³ A/cm² (@ V_{FB}-1V), and the V_{FB} shift was to -0.05 V. On the other hand, after PDA for 5 min or longer, the EOTs were



Fig. 6. Time dependence of 1000°C PDA for (a) C-V (1 MHz) and (b) J-V characteristics for HfO_xN_y thin films.



Fig. 7. Frequency dependence of C-V for the film after the 1000°C PDA for 2 min.

significantly increased to 2.1 nm or thicker, probably caused by the formation of low dielectric constant layer such as SiO_x. The remarkable reduction of the leakage current, after 1000°C PDA for 5 min or longer, also suggested the thicker interfacial layer formations. From these results, the optimum PDA condition was considered to be 1000°C for 2 min.

Figure 7 shows the frequency dependence of C-V for the film after 1000°C PDA for 2 min. Slight frequency dispersion was still observed, although the other parameters such as EOT and V_{FB} were acceptable.

Therefore, further investigation will be necessary from the device application point of view; for instance, chemical oxide layer formation prior to film deposition and/or the forming gas annealing after device fabrication.

4. CONCLUSIONS

The ECR plasma oxidation of the HfN thin films deposited on p/p^+ -Si(100) was investigated for high dielectric HfO_xN_y gate insulator formations. The EOT of 1.3 nm with the leakage current density of 4.3 x 10⁻³ A/cm² (@ V_{FB}-1V) was obtained for HfO_xN_y thin film after the 700°C HVA for 3 min followed by the 1000°C PDA for 1 min. Furthermore, the longer HVA, such as for 5 min, was found to suppress the V_{FB} shift after the 1000°C PDA. The EOT of 1.4 nm with the leakage current density of 2.3 x 10⁻³ A/cm² (@ V_{FB}-1V) was obtained for HfO_xN_y thin film after HVA for 5 min followed by 1000°C PDA for 2 min, which was the optimized annealing condition.

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