

Thermoelectric Properties of Element-Substituted CuAlO₂

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The relationships between the thermoelectric properties (α , σ , κ) of element-substituted CuAlO₂ and the thickness of its layers (conducting layer and insulating layer) were investigated. The Al-O-Al angle of the insulating layer for depends on the a-axis, and the c-axis lattice parameter is related to the thickness of the conducting layer, which it consists of O-Cu-O structures. σ and κ are related to the thickness of the conducting layer. α is related to the insulating layer. However, above $x=0.005$, the thermoelectric properties are largely influenced by impurity.

Key words: delafossite, CuAlO₂, NiO, superlattice

1. INTRODUCTION

Delafossite, whose structure is shown in Fig.1, is given by the formula ABO₂ (A=Cu, Ag, Pt, Pd; B=Al, Ga, In, Cr, Fe, Co Y, La, etc.)^[1]. Monovalent Cu or Ag (d¹⁰) for A is a semiconductor; however, Pt or Pd (d⁹) for A is known to show metallic conductivity^{[2][3]}. This observation is elucidated by of s-d₂ hybrid orbitals formed within the plane of A-ions. CuAlO₂ is known to be a p-type-conducting semiconductor. Its structure involves alternately stacking O-Cu-O and Al³⁺O₆ structures the along c-axis. The Cu⁺ ions are linearly coordinated by two O²⁻ ions to form a O-Cu-O dumbbell along the direction parallel to the c-axis, whereas, Al³⁺ ions are octahedral-coordinated by six O²⁻ ions^{[4][5]}. The study focuses on the relationship between the thickness of the layer and the thermoelectric properties of element-substituted CuAlO₂. The possibility of realizing a high-efficiency thermoelectric conversion material that possesses a layer structure was reported^{[6][7]}. In addition, various materials were reported^[8-13]. It is of interest that the layer structure of CuAlO₂ exhibits a large thermopower. However, it has a very low electrical conductivity ($\sim 1 \times 10^2 \text{ Sm}^{-1}$ at 873K)^{[14][15]}. If electrical conductivity can be improved, thermoelectric efficiency is expected to be high. However, reports on the thermoelectric properties of element-substituted CuAlO₂ have been few; in addition there has thus far been no report on thermal conductivity.

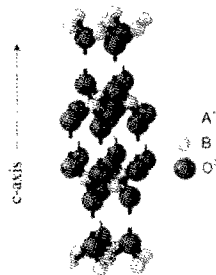


Fig.1 Crystal structure of delafossite-type ABO₂.

The purpose of this study is to determine the relationships between the thermoelectric properties of element-substituted CuAlO₂ and the thickness of its layers (thickness of conducting layer and insulating layer).

2. EXPERIMENTAL PROCEDURE

CuO, Al₂O₃ and NiO powders were used as raw materials for solid-state reaction. These materials were mixed in a molar ratio of Al₂O₃: NiO= (1-x): x (x=0, 0.001, 0.0025, 0.005 and 0.0075) a planetary ball mill for 2 h. The mixed powders were calcined at 1373 K for 12 h. Then, CuO was mixed with the calcined powder, ball-milled for 24 h, palletized, and calcined at 1373 K for 24 h. The sintered pellet was pulverized. Then, it was milled for 24 h. The mixture was pressed at 245 MPa and sintered at 1473 K for 48 h. Electrical conductivity and Seebeck coefficient were measured by the four-probe method, and thermal conductivity was measured by the laser flash method. Phase, crystal structures and lattice parameters were analyzed by X-ray diffraction (XRD) analysis.

3. RESULTS AND DISCUSSION

Figure 2 shows changing of a-axis and c-axis parameters for CuAl_{1-x}Ni_xO₂ samples lattice parameter parameters for CuAl_{1-x}Ni_xO₂ samples. The a-axis drastically decreased at $x=0.001$ and in increased at $x=0.0025$; on the other hand, the c-axis increased at $x=0.001$ and decreased extremely at $x=0.0025$. The drastic decrease in the a-axis reduced Al-O-Al angle. Then, the angle makes the insulating layer thick; meanwhile, the increase in c-axis gives rise to the increase in the O-Cu-O distance of the conducting layer. Beyond $x=0.001$, an increase in the a-axis causes Al-O-Al angle leading to expand to the insulating layer becoming thin. Furthermore, the extreme decreasing of c-axis at $x=0.0025$ reduces the O-Cu-O distance of the conducting layer. According to XRD results, an impurity phase such as NiAl₂O₄ appeared at $x=0.0075$. From this

fact, it is suggested that the Ni system compound exists inside the conducting layer. If the Ni system compound is larger than the distance between layers of O-Cu-O, the a-axis increases, and the c-axis decreases. As a result, it is suggested that only the thickness of the insulating layer decreases.

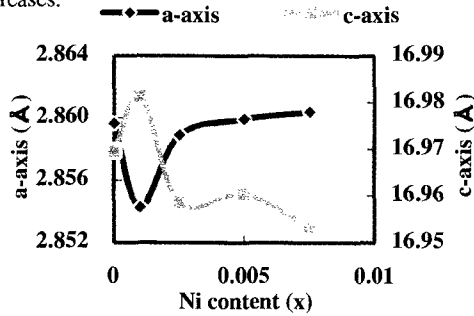


Fig.2 Changing of a-axis and c-axis parameters for $\text{CuAl}_{1-x}\text{Ni}_x\text{O}_2$ samples.

Figure.3 shows the relationship between lattice parameter and Seebeck coefficient (α) for $\text{CuAl}_{1-x}\text{Ni}_x\text{O}_2$ samples. (a) and (b) are α of RT and 850K, respectively. The α at RT decreases with increasing c-axis at $x=0.001$. It is suggested that the thickness of insulating layer is larger than thickness of conduction layer. α increases with decreasing c-axis at $x=0.0025$. Above $x=0.0025$, it decreases with decreasing c-axis. It is suggested that α decrease under the influence of the Ni system compounds impurity.

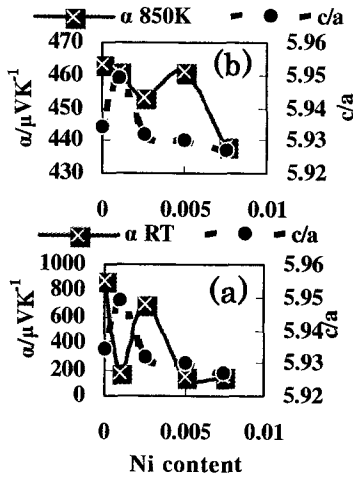


Fig.3 Relationship lattice parameter and α for $\text{CuAl}_{1-x}\text{Ni}_x\text{O}_2$ samples. (a) and (b) are α of RT and 850K, respectively.

Figure 4 shows the relationship between lattice parameter and electrical conductivity (σ) for $\text{CuAl}_{1-x}\text{Ni}_x\text{O}_2$ samples. (a) and (b) are σ of RT and 850K, respectively. σ shows a behavior similar to that of c-axis. Thus, it is a related to the in thickness of the conducting layer. Above $x=0.005$ at 850K, σ is larger than its value at $x=0.0025$. The increasing of σ at

850K is suggested to be attributable to the existence of Ni system compounds impurity.

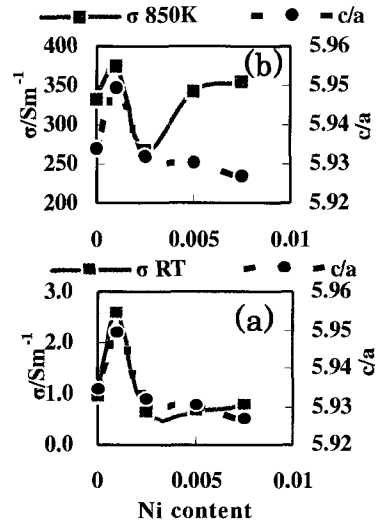


Fig.4 Relationship between lattice parameter and σ for $\text{CuAl}_{1-x}\text{Ni}_x\text{O}_2$ samples. (a) and (b) are σ of RT and 850K, respectively.

Figure 5 shows the relationship between lattice parameter and thermal conductivity (κ) for $\text{CuAl}_{1-x}\text{Ni}_x\text{O}_2$ samples. (a) and (b) are κ of RT and 850K, respectively. κ shows the same behavior as a-axis. Thus, it is related to the thickness of the conducting layer. Above $x=0.0075$ at 850K, the κ is larger than its value at $x=0.005$. The increasing of κ at 850K is suggested to be due to the influence of Ni system compounds impurity.

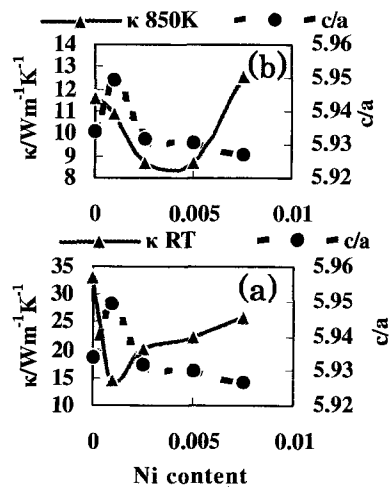


Fig.5 Relationship between lattice parameter and κ for $\text{CuAl}_{1-x}\text{Ni}_x\text{O}_2$ samples. (a) and (b) are κ of RT and 850K, respectively.

4. CONCLUSION

The relationships between layer structure and thermoelectric properties (α , σ , κ) were determined. a-axis parameter changes with the Al-O-Al angle of the insulating layer of, and c-axis parameter changes with the O-Cu-O length of the conducting layer. σ and κ are related to the conducting-layer thickness. α is related to the insulating-layer thickness. However, above $x=0.005$, thermoelectric properties become largely influenced by Ni compounds impurity.

5. REFERENCE

- [1] T. Koyanagi, H. Harima, A. Yanase and H. Katayama-Yoshida, *J. Phys. Chem. Solid* 64 1443-1446(2003).
- [2] K. Koumoto, H. Koduka and W. S. Seo, *Proceeding of 19th Int. Conf. Thermoelectrics* 232-235(2000).
- [3] K. Koumoto, H. Koduka, and W. S. Seo, *J. Mater. Chem.* 11, 251-252 (2001).
- [4] H. Kawazoe, M. Yasukawa, H. Hyoudo, M. Kurita, H. Yanagi and H. Hosono, *Nature (London)* 389, 939 (1997).
- [5] B. J. Ingram and T. O. Mason, R. Asahi, K. T. Park and A. J. Freeman, *Phys. Rev. B.*, 64, 155114-1, 7 (2001).
- [6] L. D. Hicks, M. S. Dresselhaus, *Phys. Rev. B*47, 12727-12731 (1993).
- [7] L. D. Hicks, M. S. Dresselhaus, *Phys. Rev. B*47, 16631-16634 (1993).
- [8] I. Terasaki, Y. Sasago, K. Uchinokura, *Phys. Rev. B* 56, R12 685-687 (1997).
- [9] I. Matsubara, R. Funahashi, *Ceramics*, 36, 8, 582-584 [in Japanese] (2001).
- [10] H. Yakabe, K. Kikuchi, I. Terasaki, Y. Sasago, K. Uchinokura, *Proceeding 16th Int. Conf. Thermoelectronics*, 522-527 (1997).
- [11] S. Li, R. Funahashi, I. Matsubara, K. Ueno, S. Sodeoka, H. Yamada, *Proceeding 18th Int. Conf. Thermoelectronics*, 581-583 (1999).
- [12] S. Hirano, S. Isobe, T. Tani, N. Kitamura, I. Matsubara, K. Koumoto, *Proceeding Thermoelectric Conversion Symposium (Thermoelectric Conversion Research Committee of Japan)*, 68-69(2002).
- [13] M. Ohta, M. Kazeoka, W. S. Seo, K. Koumoto, *Proceeding 17th Int. Conf. Thermoelectronics*, 606-609 (1998).
- [14] T. Kurotori, S. Sugihara, *The 43rd Symposium on Basic Science of Ceramics Abstract Book* 24-25(2005).
- [15] T. Kurotori, S. Sugihara, *Mater. Trans. Vol.46*, 7 1462-1465(2005).

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